

X-ray Diffraction studies of irradiated Materials at BNL Experimental Facilities - N. Simos (Oct. 9, 2014)

MATERIALS:

Graphite polymorphs, h-BN, Be, AlBeMet, Tungsten, Molybdenum, Glidcop, Mo-Gr, Cu-CD, carbon fiber composites, superalloys (Ti6Al4V, s-INVAR and gum metal) and metal-metal interfaces

Irradiations:

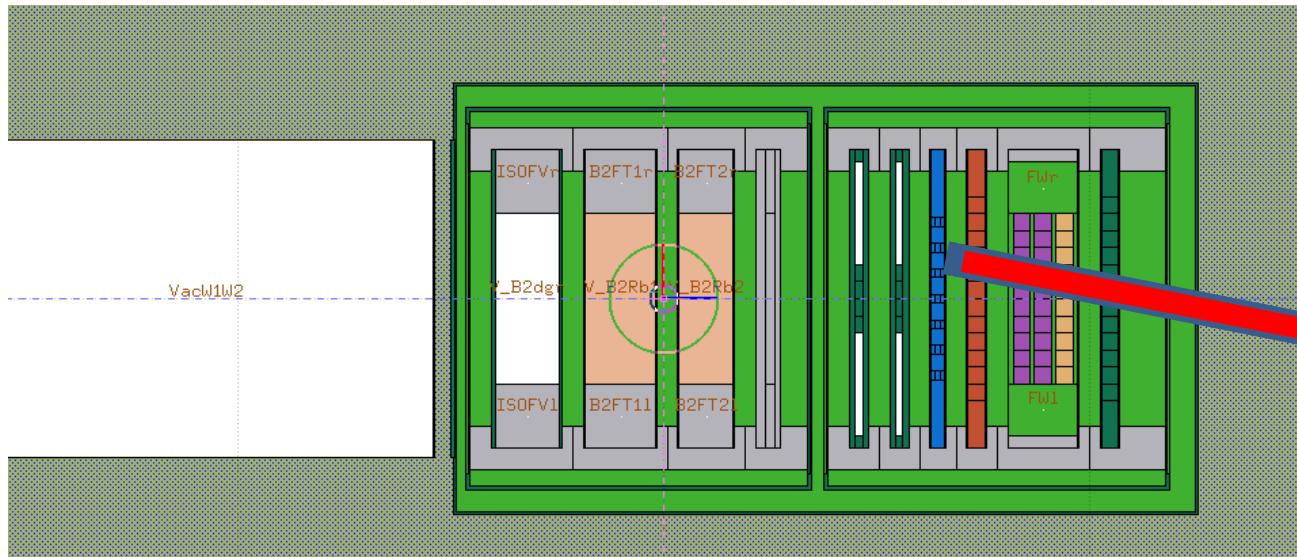
- 118-200 MeV Protons at BNL BLIP***
- Fast Neutrons at BNL BLIP***
- 28 MeV Protons at Tandem***
- Neutrons at Tandem (low temperature)***

X-ray Studies (completed)

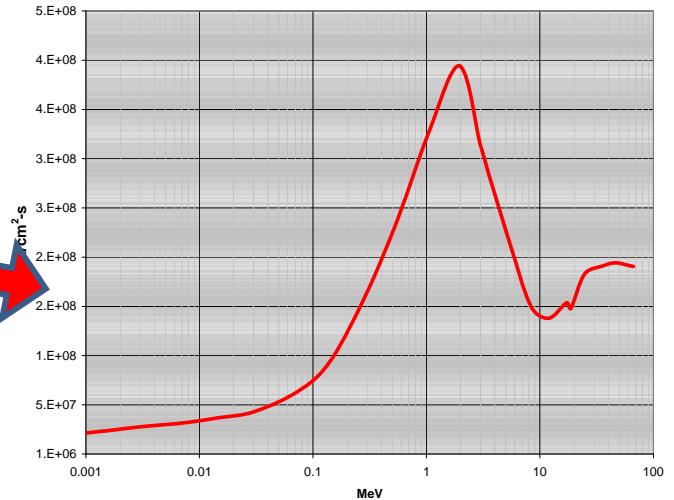
- (a) using monochromatic high energy X-rays
- (B) high energy x-rays EDXRD (Phase I & Phase II)

MICROSCOPY (at CFN): SEM/EDS, annealing, DSC and TG/DTA

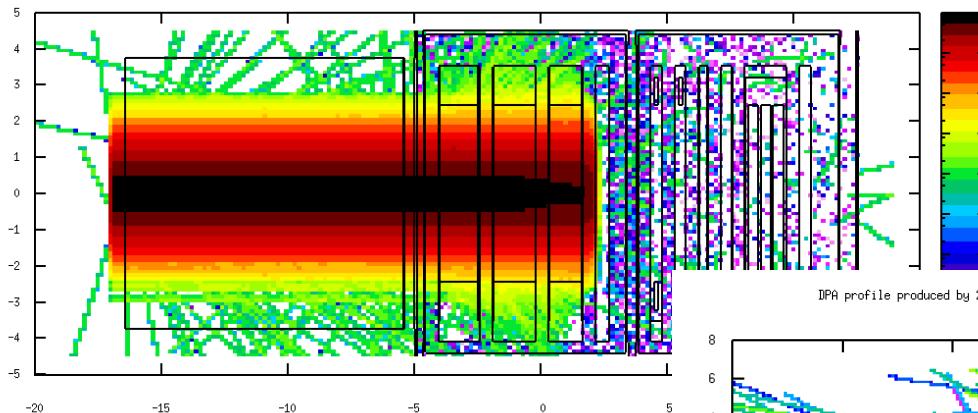
Spallation Neutron Irradiation at BLIP



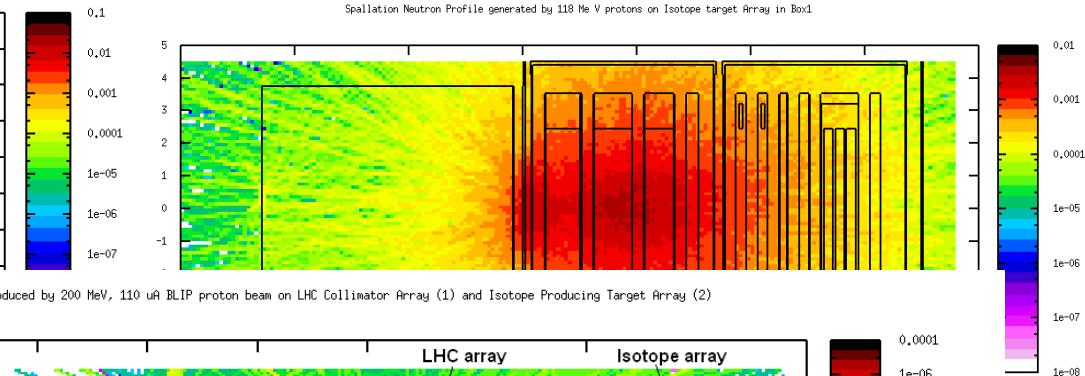
n_spectra at BLIP target station irradiating nanostructured coatings
graph is for normalized proton flux of 10^{12} p/s



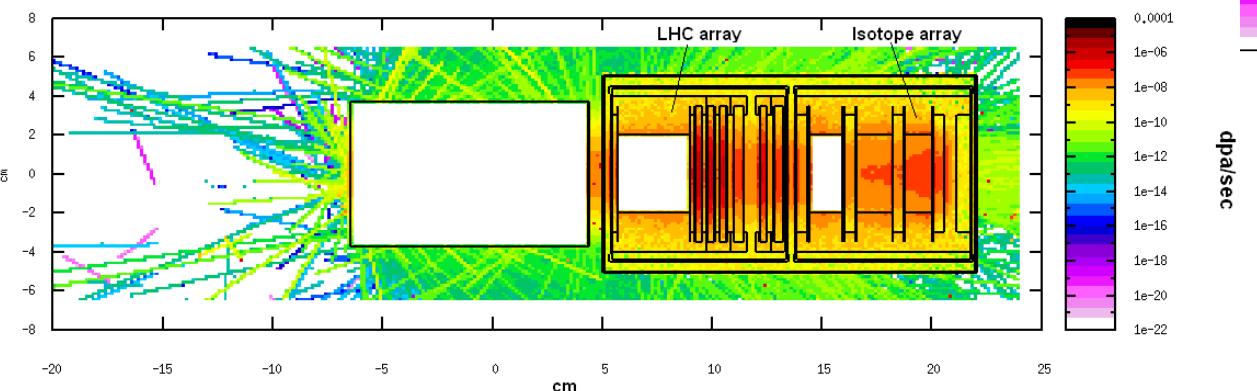
Proton Distribution Profile - 118 MeV BLIP Proton Beam with Isotope targets in Box-1



Spallation Neutron Profile generated by 118 MeV protons on Isotope target Array in Box1

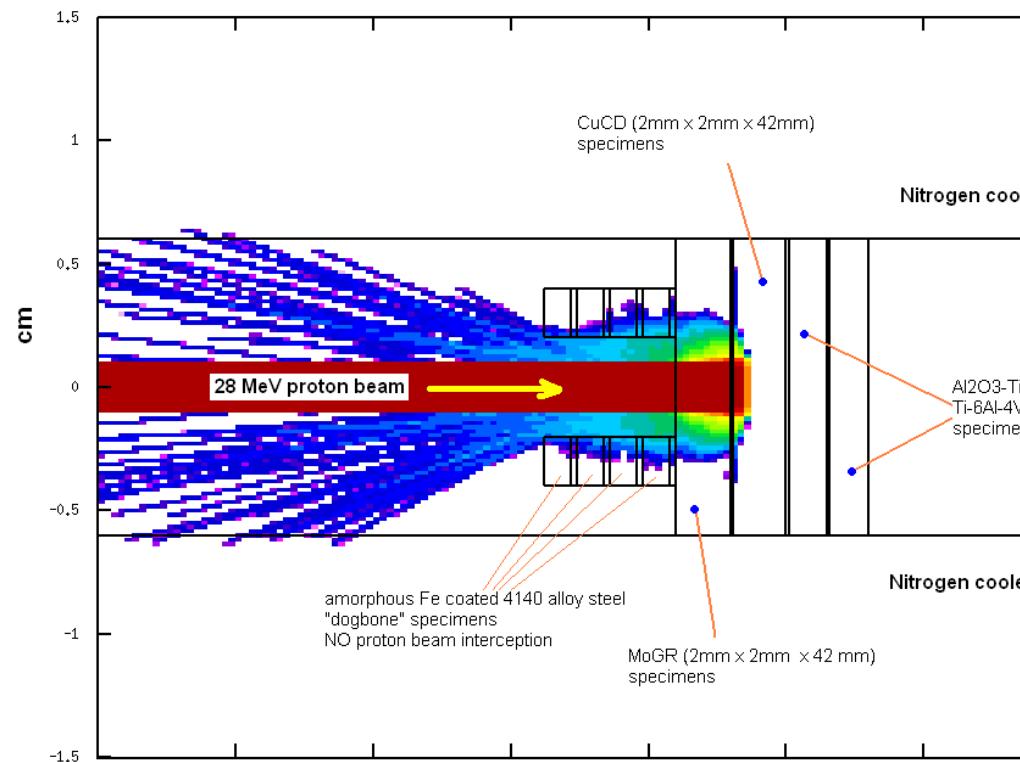


DPA profile produced by 200 MeV, 110 uA BLIP proton beam on LHC Collimator Array (1) and Isotope Producing Target Array (2)

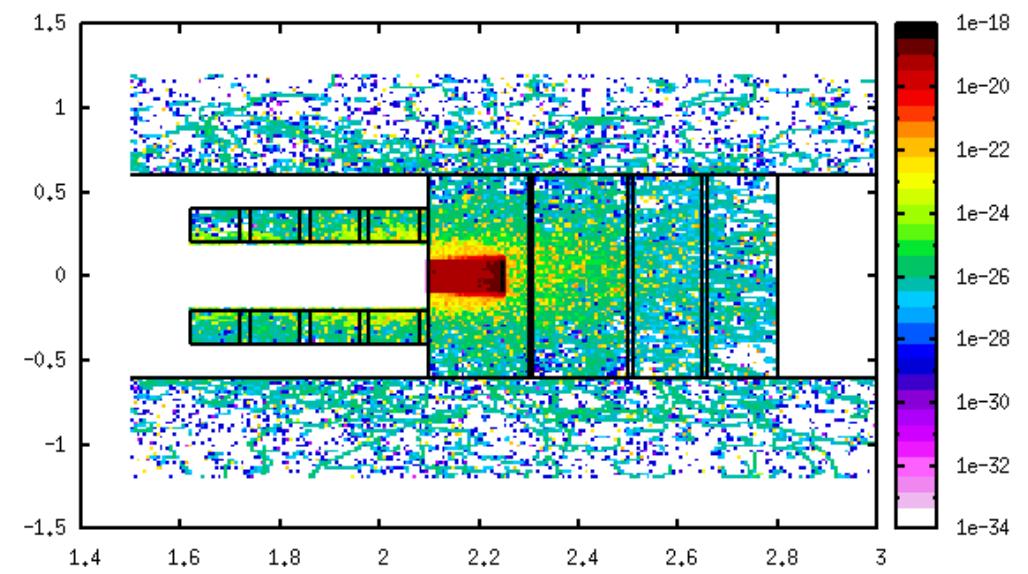
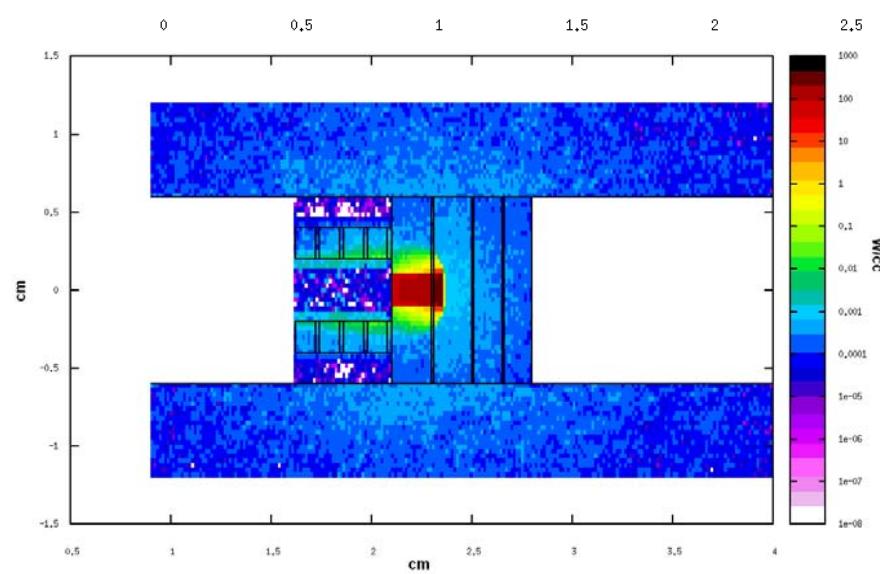
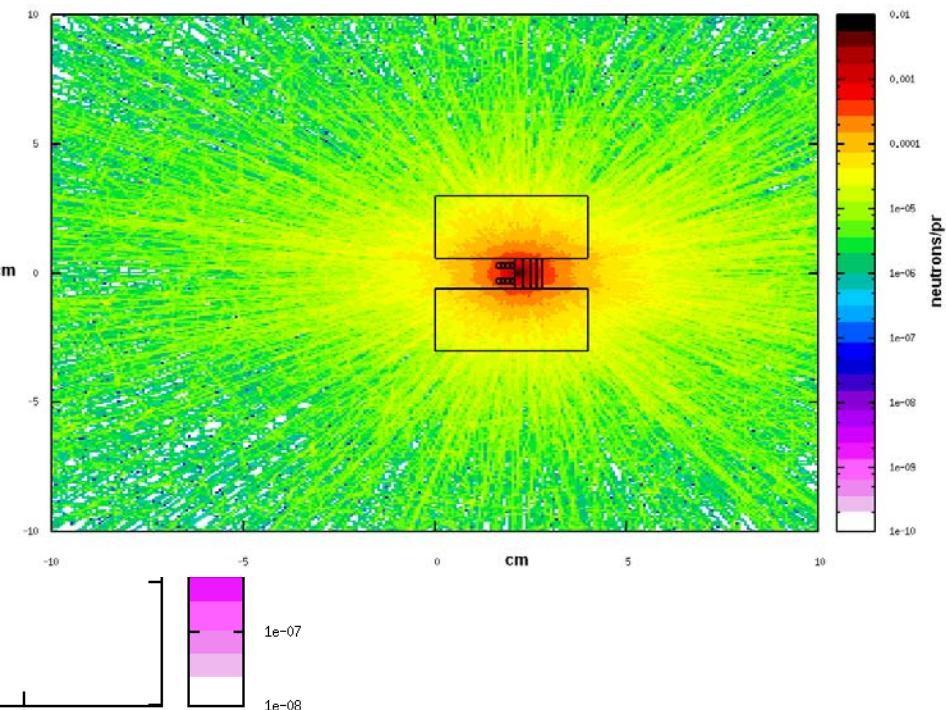


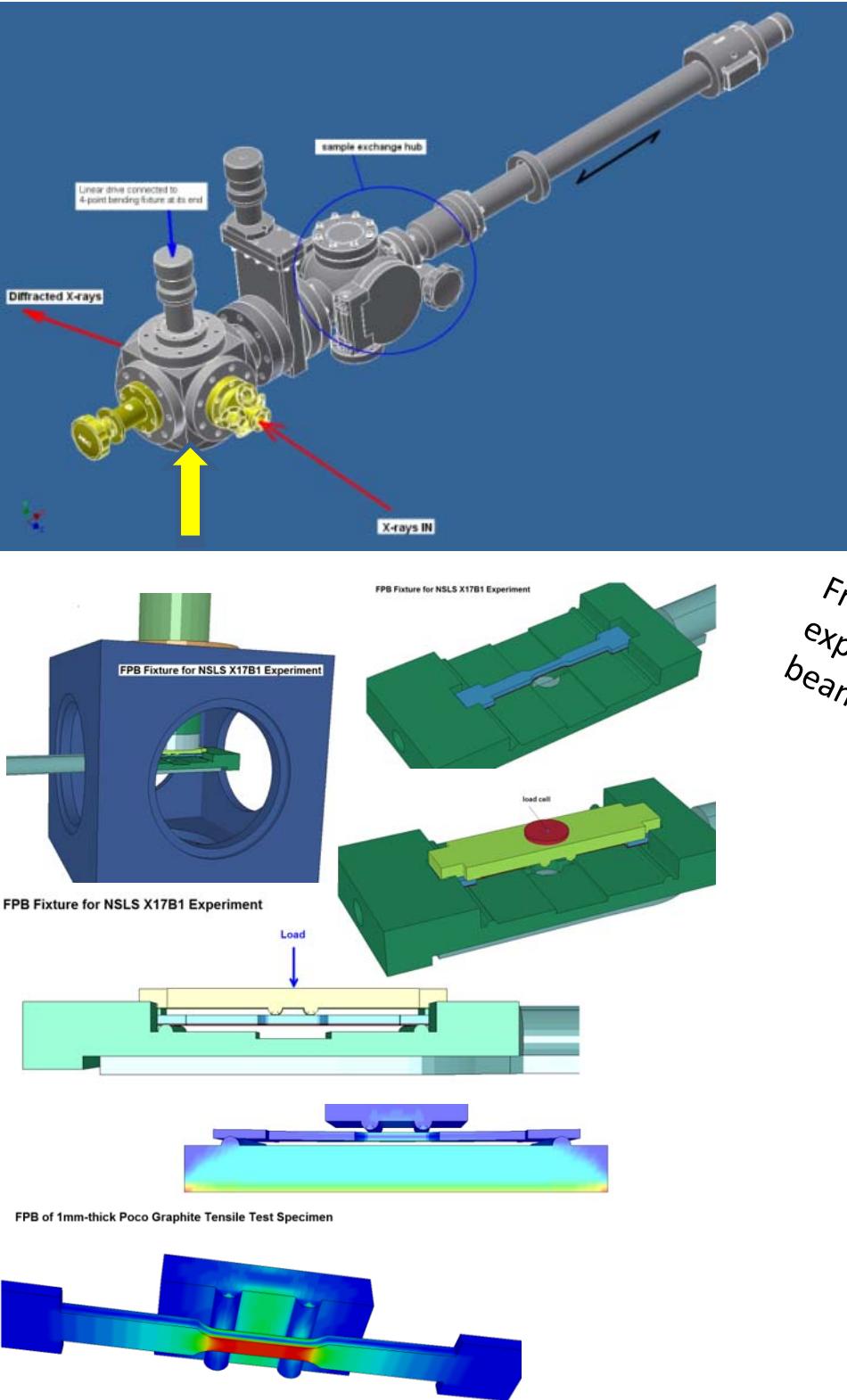
28 MeV Proton Irradiation at Tandem

Localized Damage Followed by EDXRD Studies



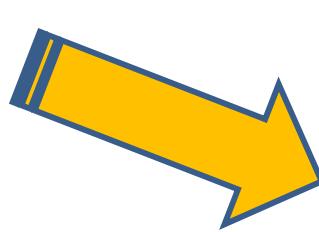
Generated Neutron Profile inside the ORTEC Vacuum Chamber



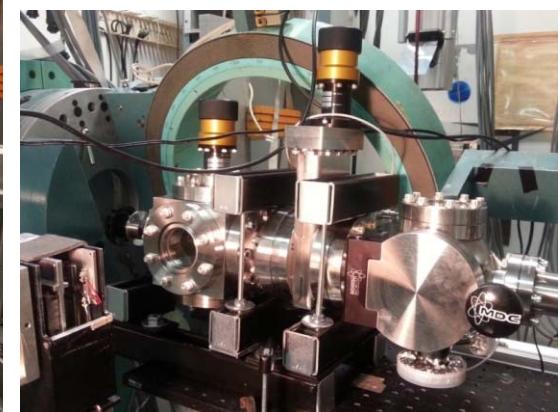
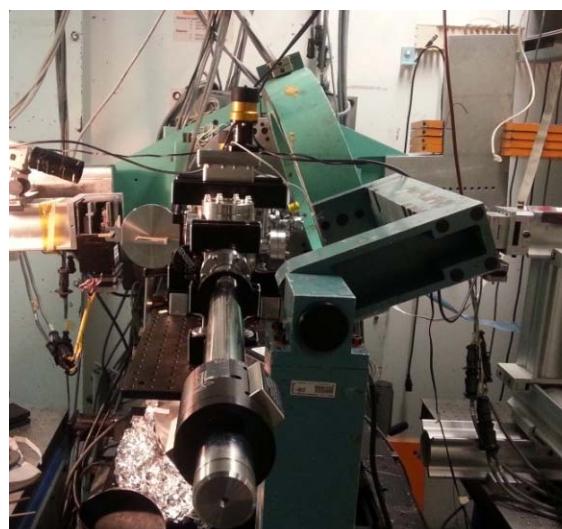
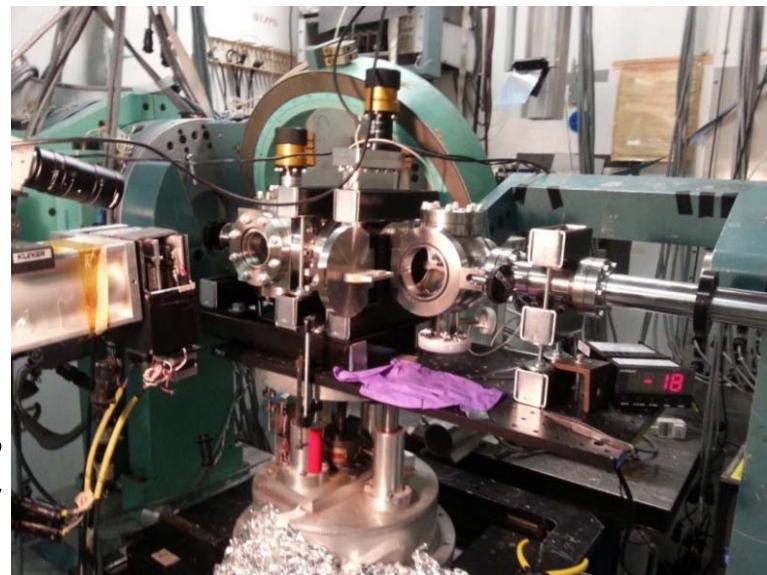


Multi-functional stage capable of handling

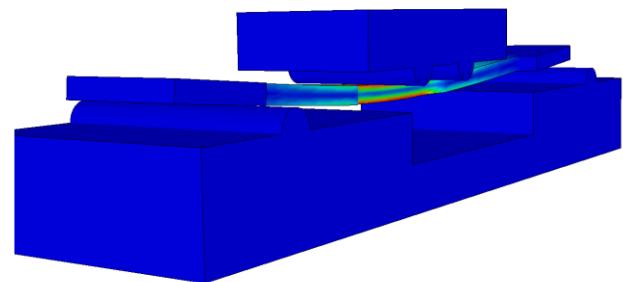
Real size irradiated specimens, under vacuum and four point bending state of stress
and eventually
Heating/annealing via a portable, collimated laser beam
Tensile stress-strain test



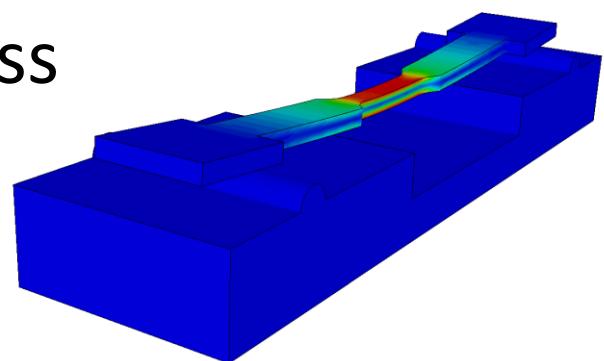
From concept to a versatile experimental stage at X17B1 beamline at NSLS



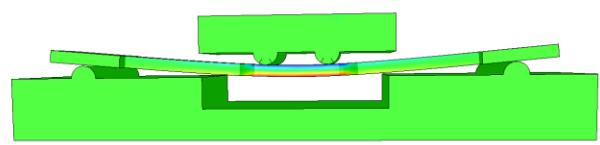
von Mises Stress



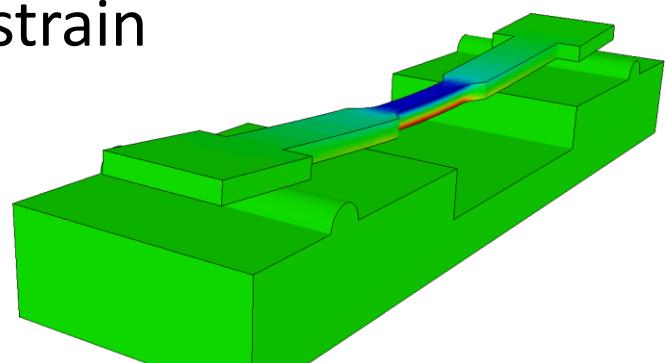
stress



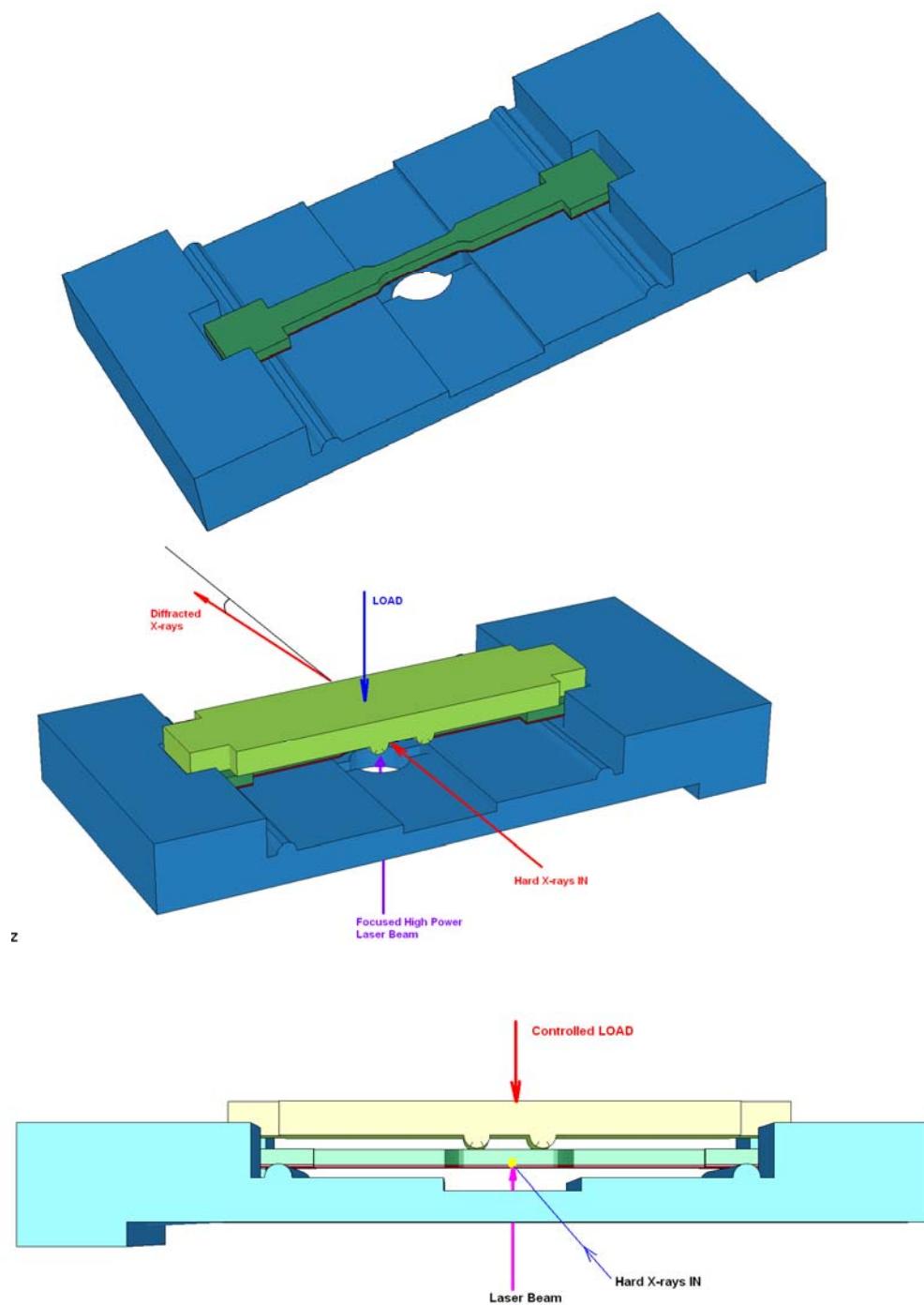
Plastic Strain

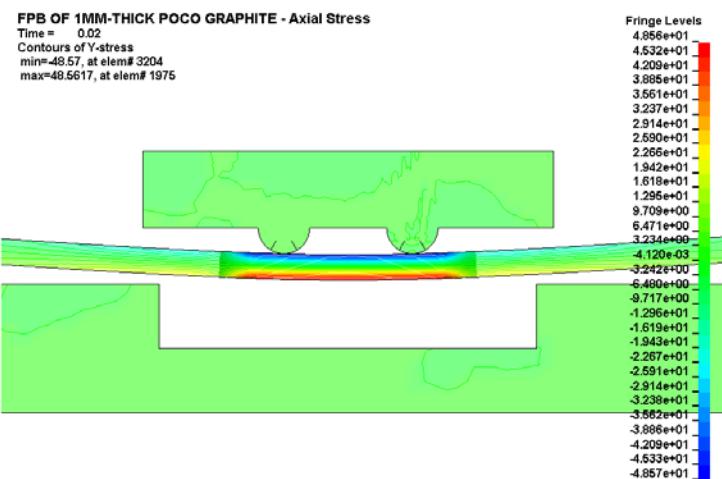
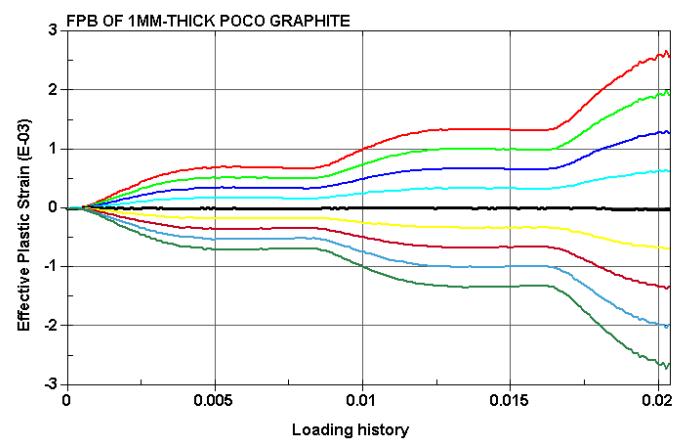
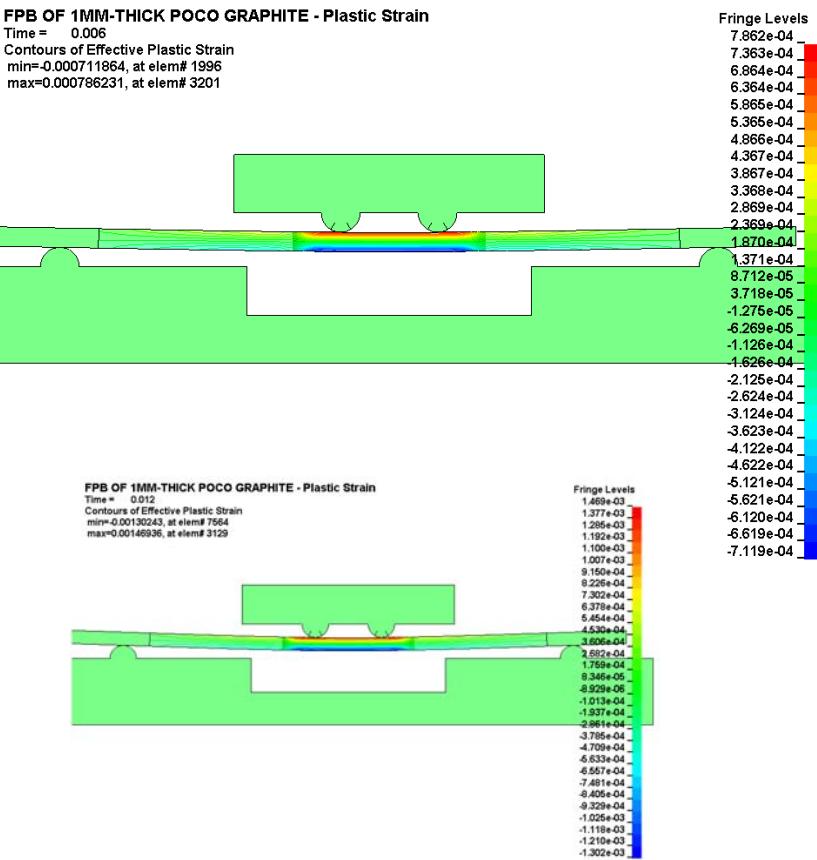
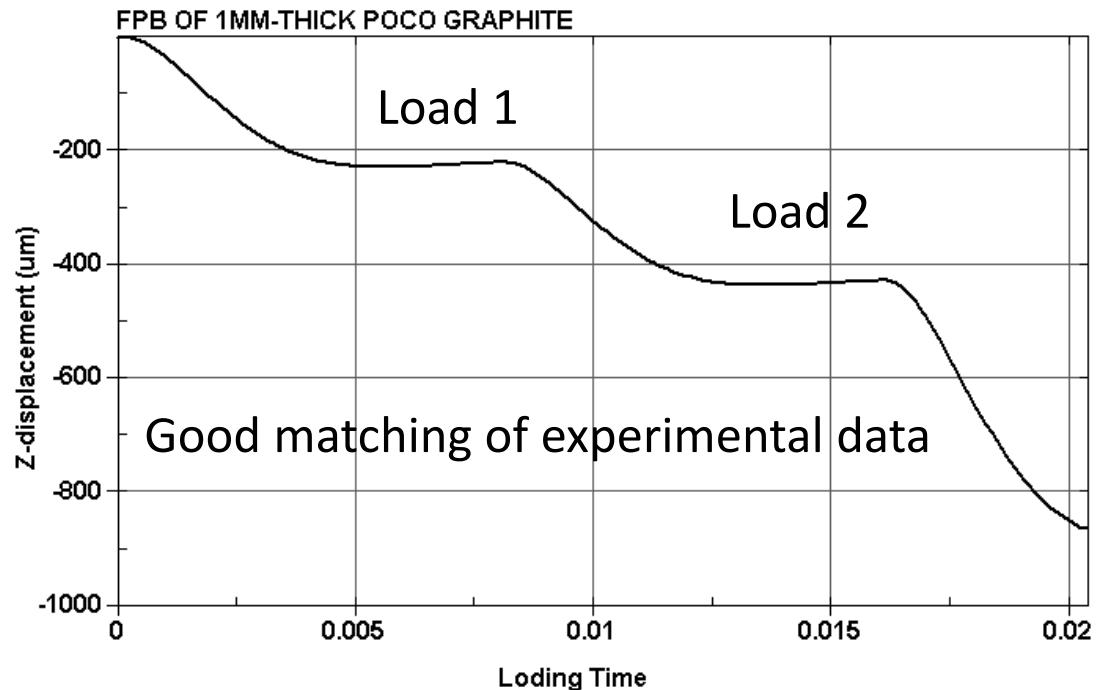


strain



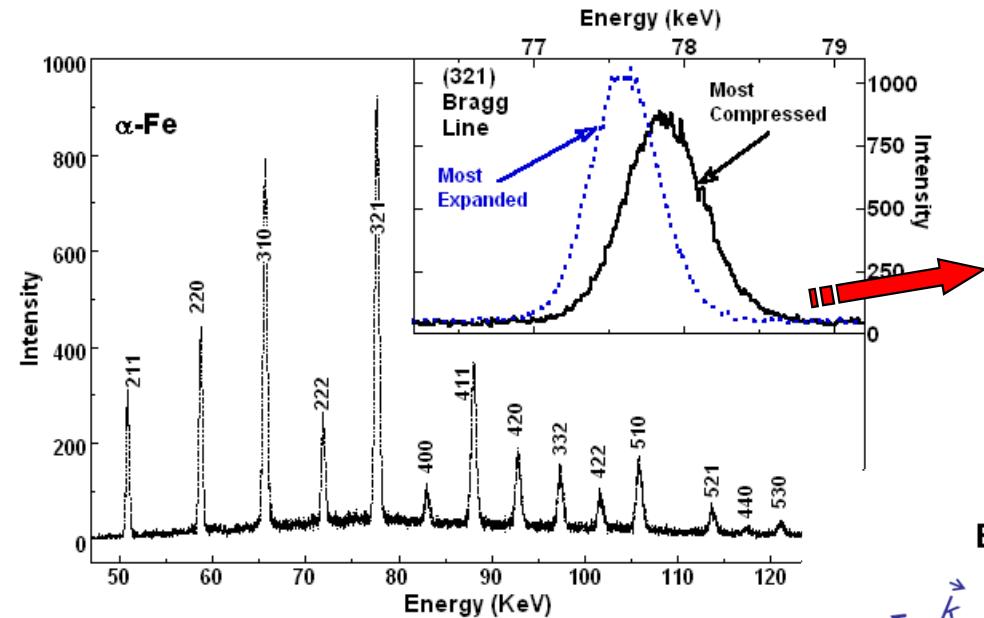
z





STRAIN MAPPING

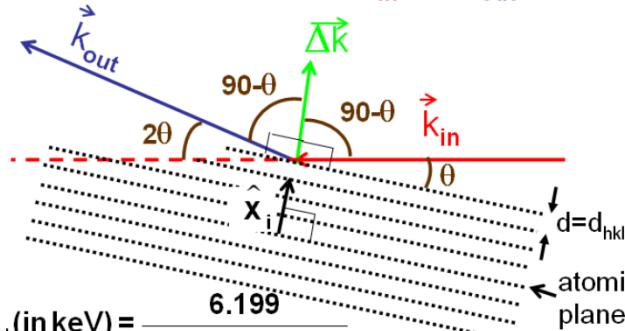
Energy Dispersive Diffraction Mode



$$E_{hkl} \text{ [in keV]} = \frac{6.199}{d_{hkl} \sin \theta} \quad \rightarrow \quad \varepsilon = \frac{\Delta d}{d_0}$$

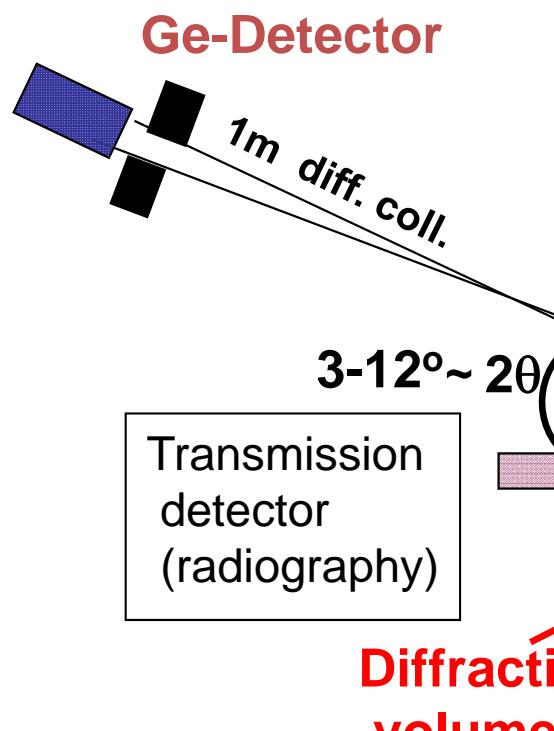
Like having imbedded inter-atomic strain gauges !!!!

$$E = h \frac{c}{\lambda} = \frac{h}{2\pi} k \quad |k_{in}| = |k_{out}| = k$$

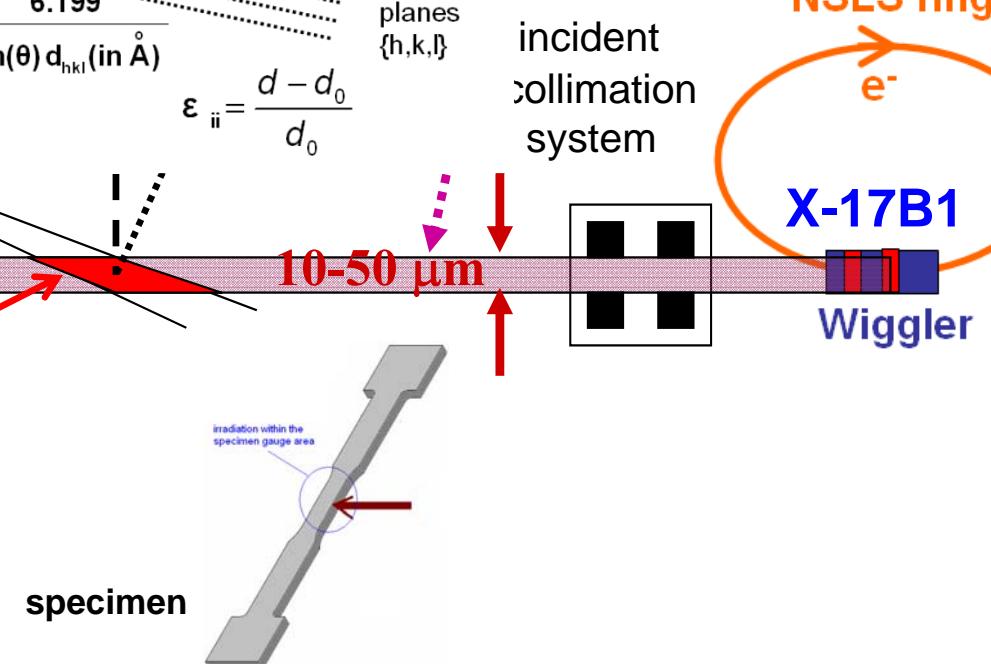


$$E_{hkl} \text{ (in keV)} = \frac{6.199}{\sin(\theta) d_{hkl} \text{ (in } \text{\AA})}$$

$$\varepsilon_{||} = \frac{d - d_0}{d_0}$$

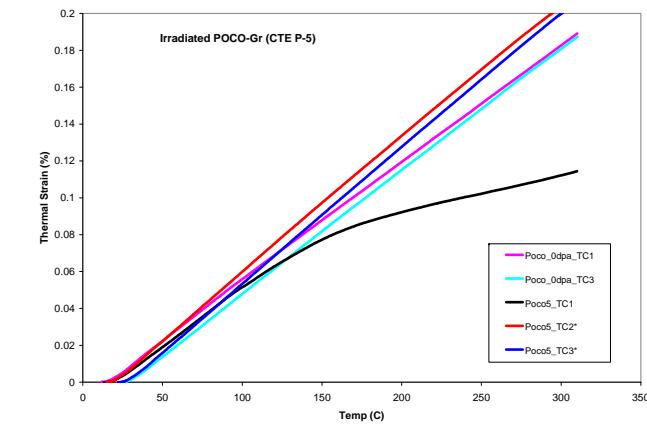
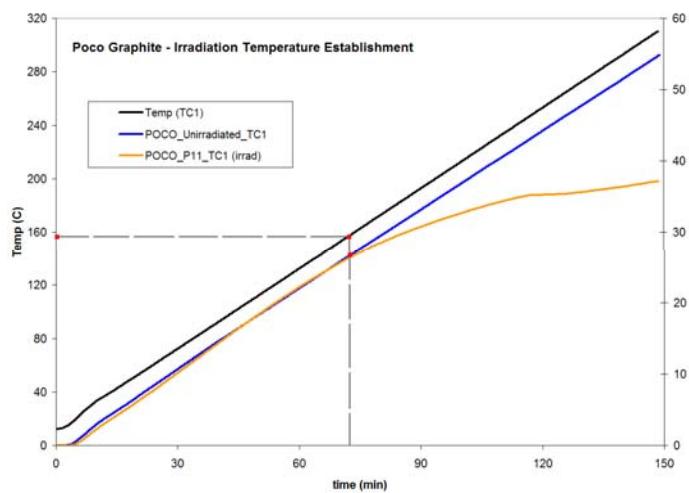
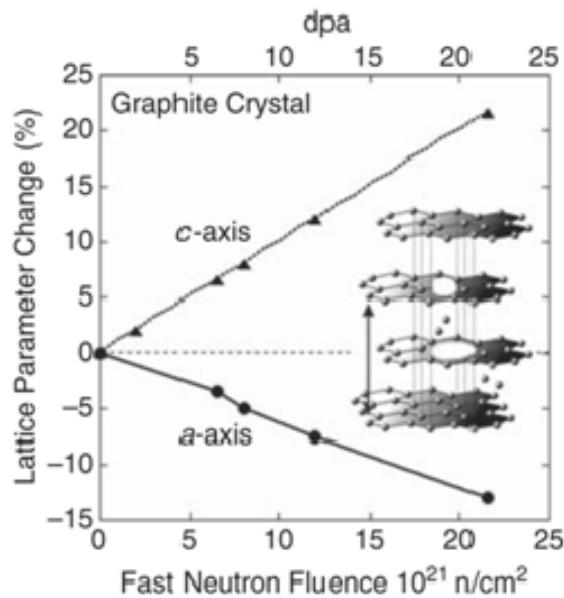


: “White Beam”

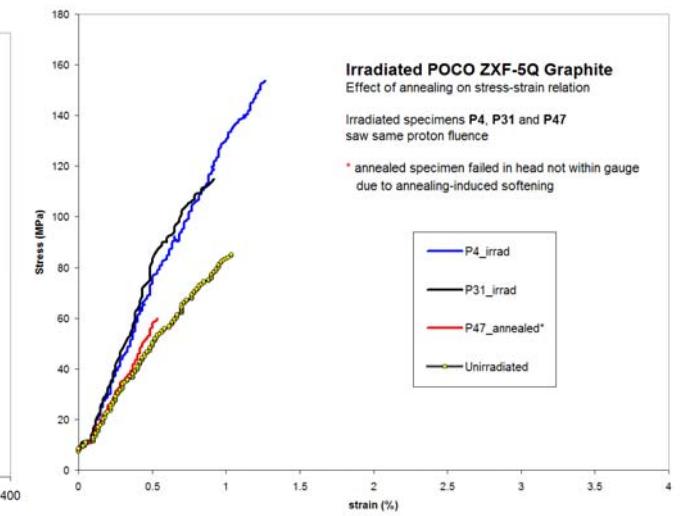
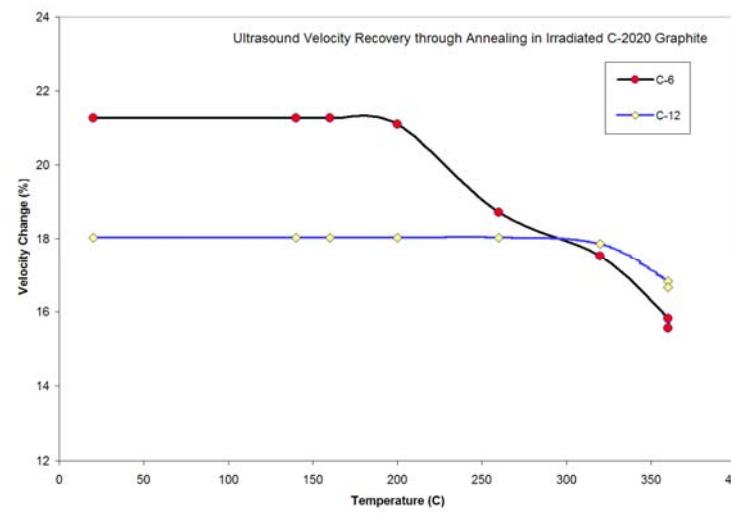
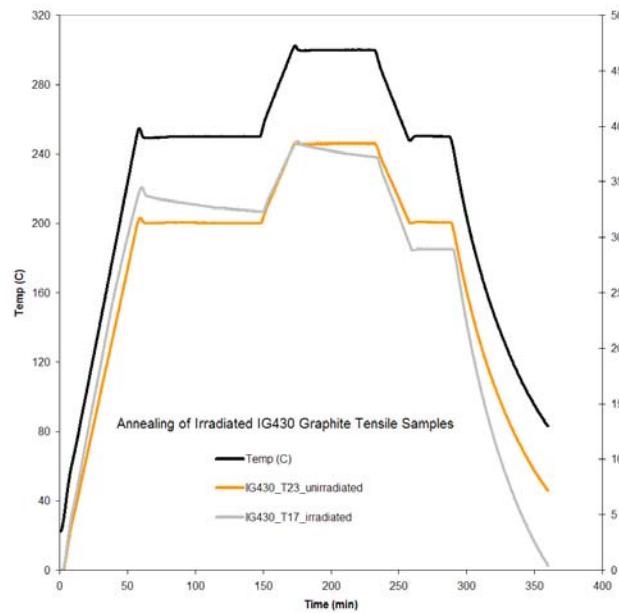


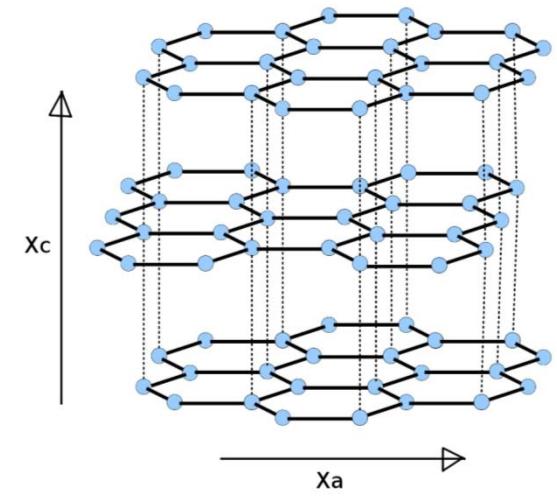
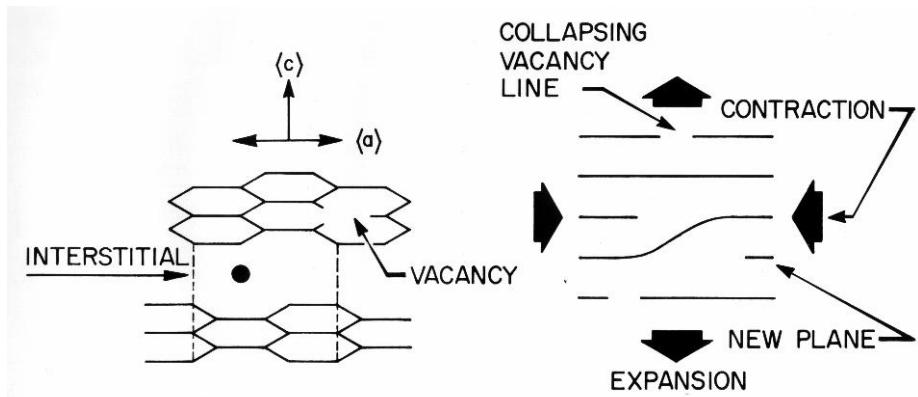
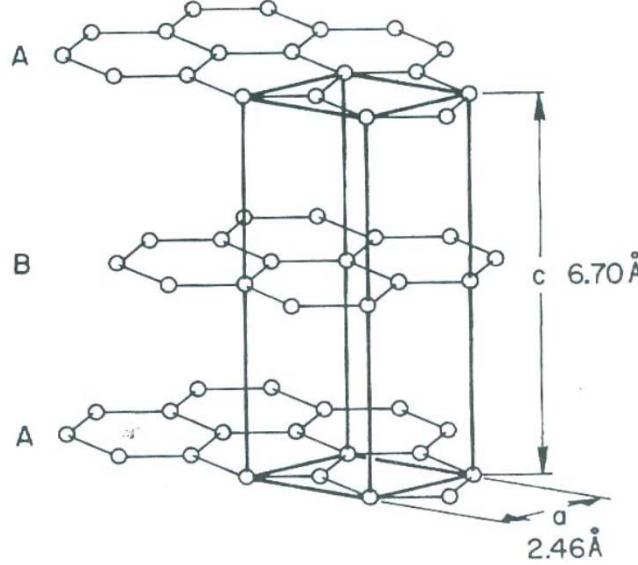
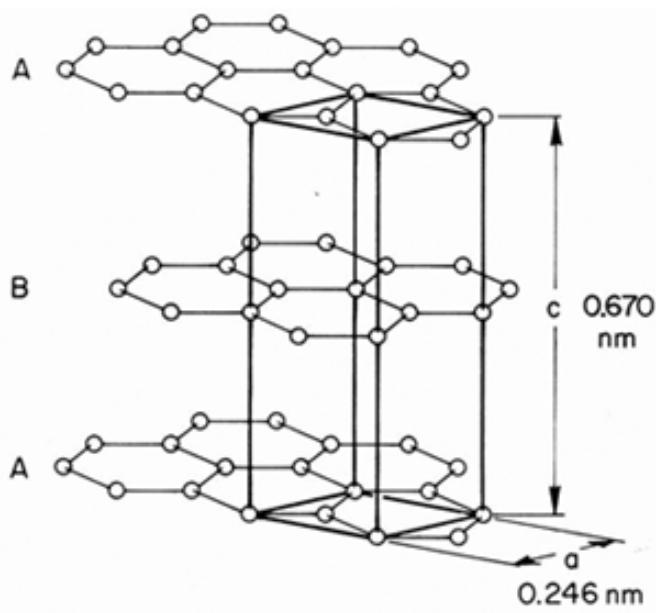
Graphite

Important to know what occurs during irradiation and post-irradiation annealing (mobilization of interstitials/vacancies)



This is what we observe in BULK
What happens at the crystal level?
How is E is affected or is strain in crystal related to bulk?



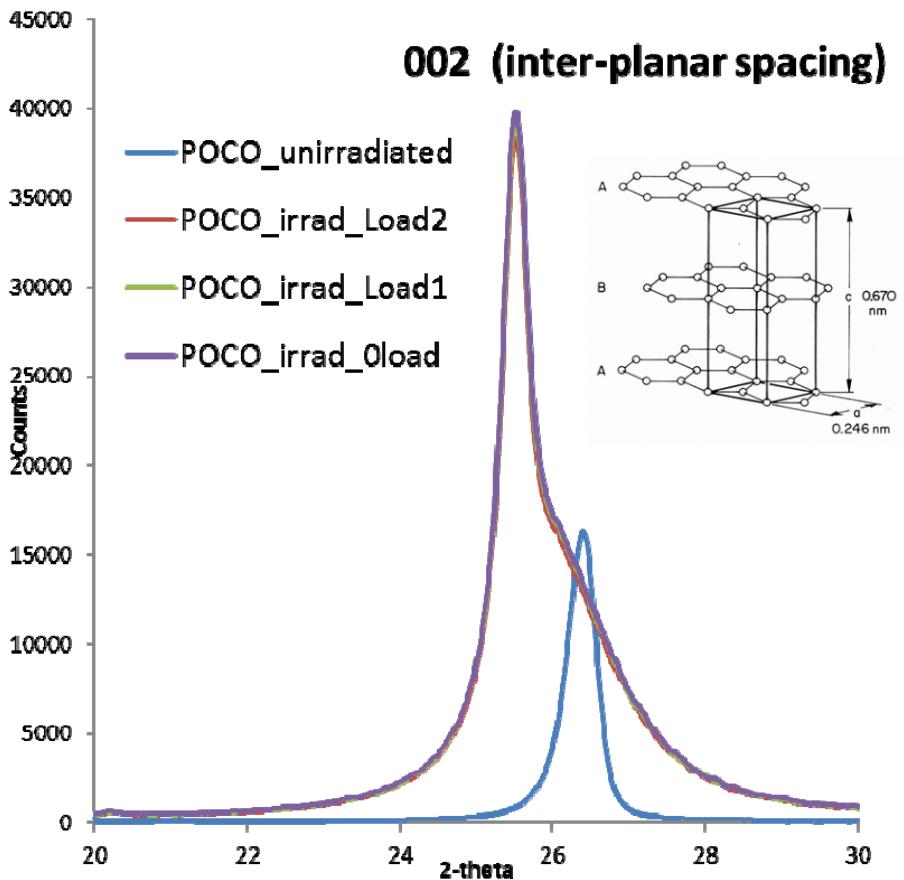
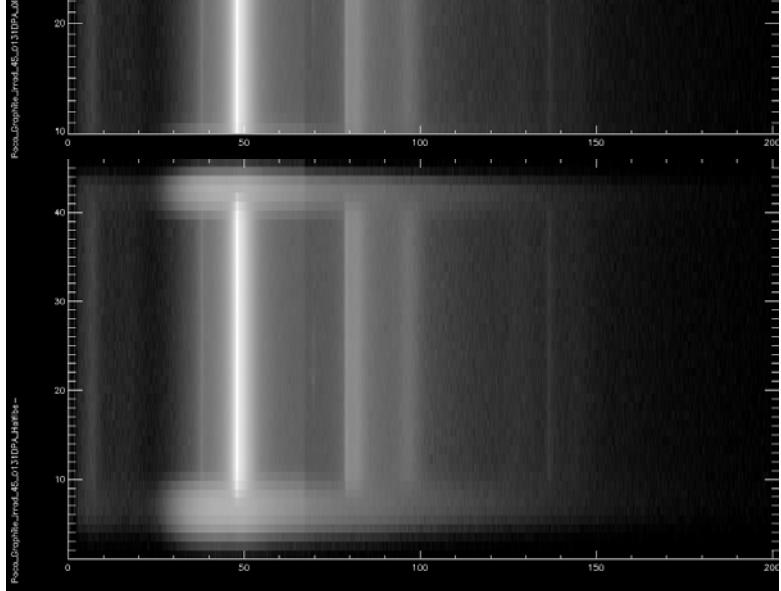
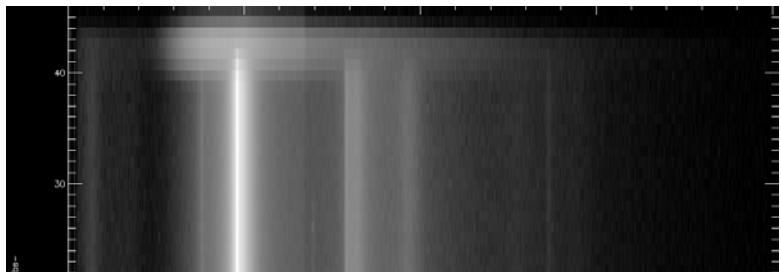
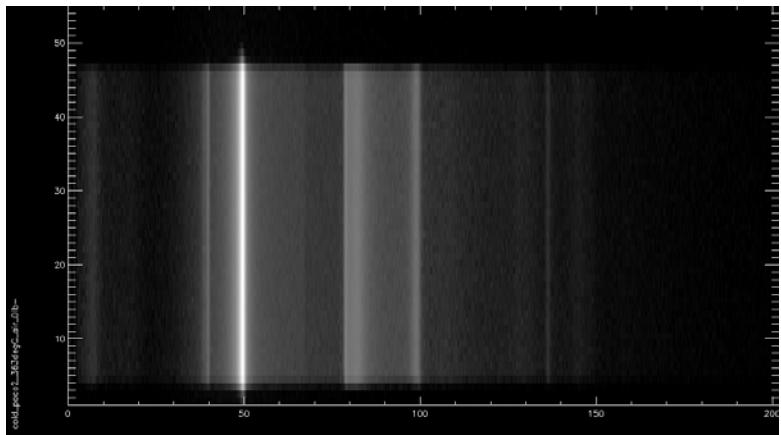


Interstitial defects will cause crystallite **growth** perpendicular to the layer planes (c-axis direction)

Coalescence of vacancies will cause a **shrinkage** parallel to the layer planes (a-axis direction)

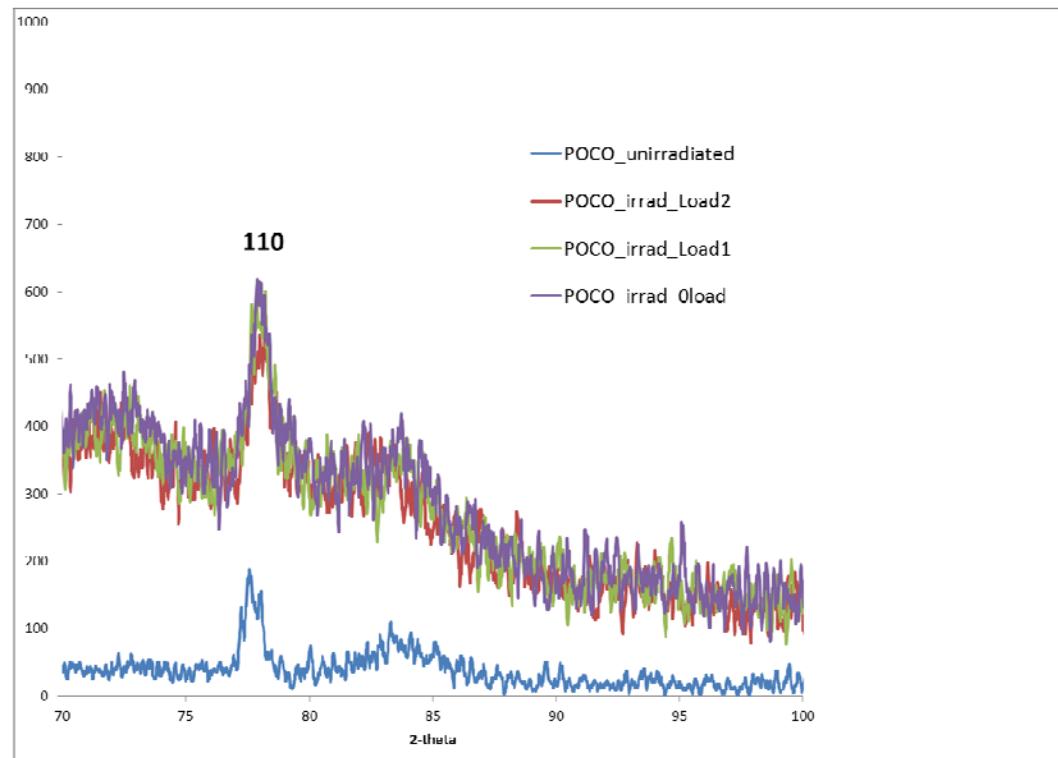
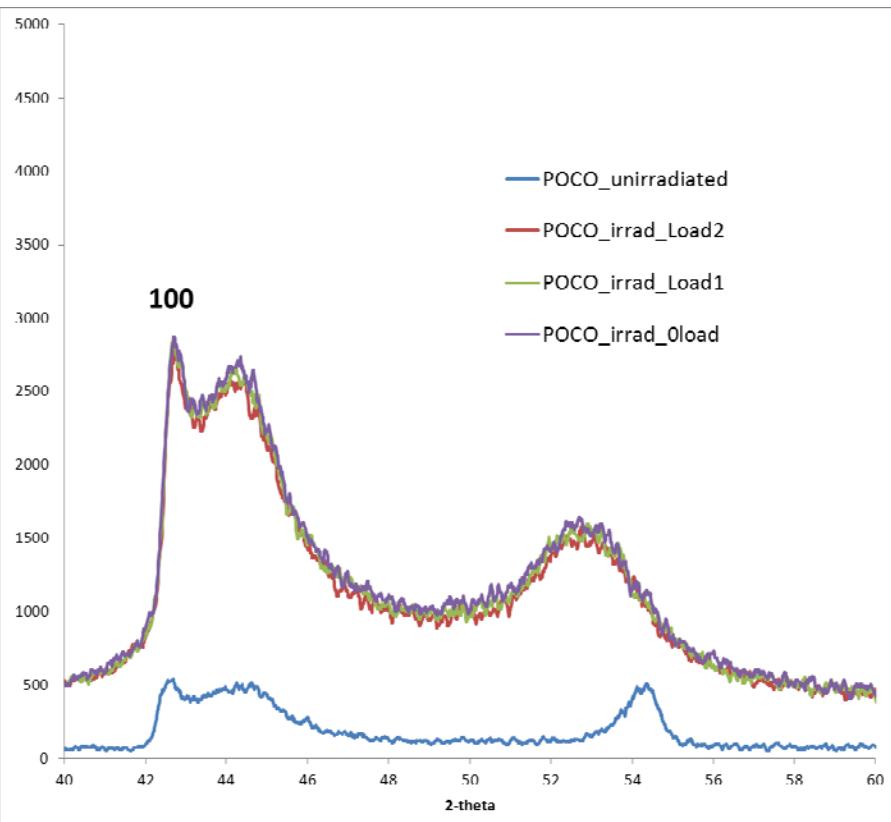
Graphite

Various grades, including Carbon fiber composites under different irradiations



This 002 peak also broadens asymmetrically, with a bias towards smaller angles indicating an increase in average interlayer distance.

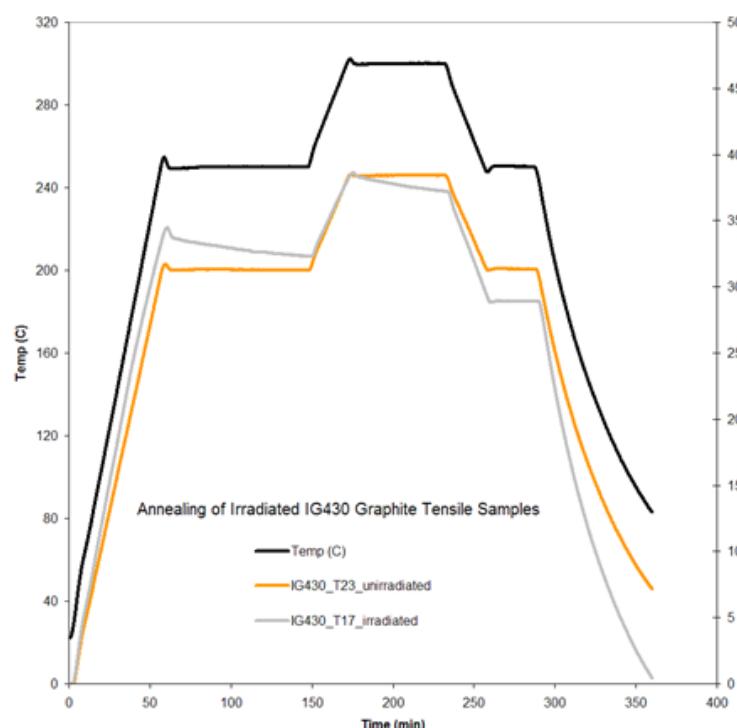
The (002) diffraction spot also broadens in single crystal images, suggesting a range of values for the interlayer distance

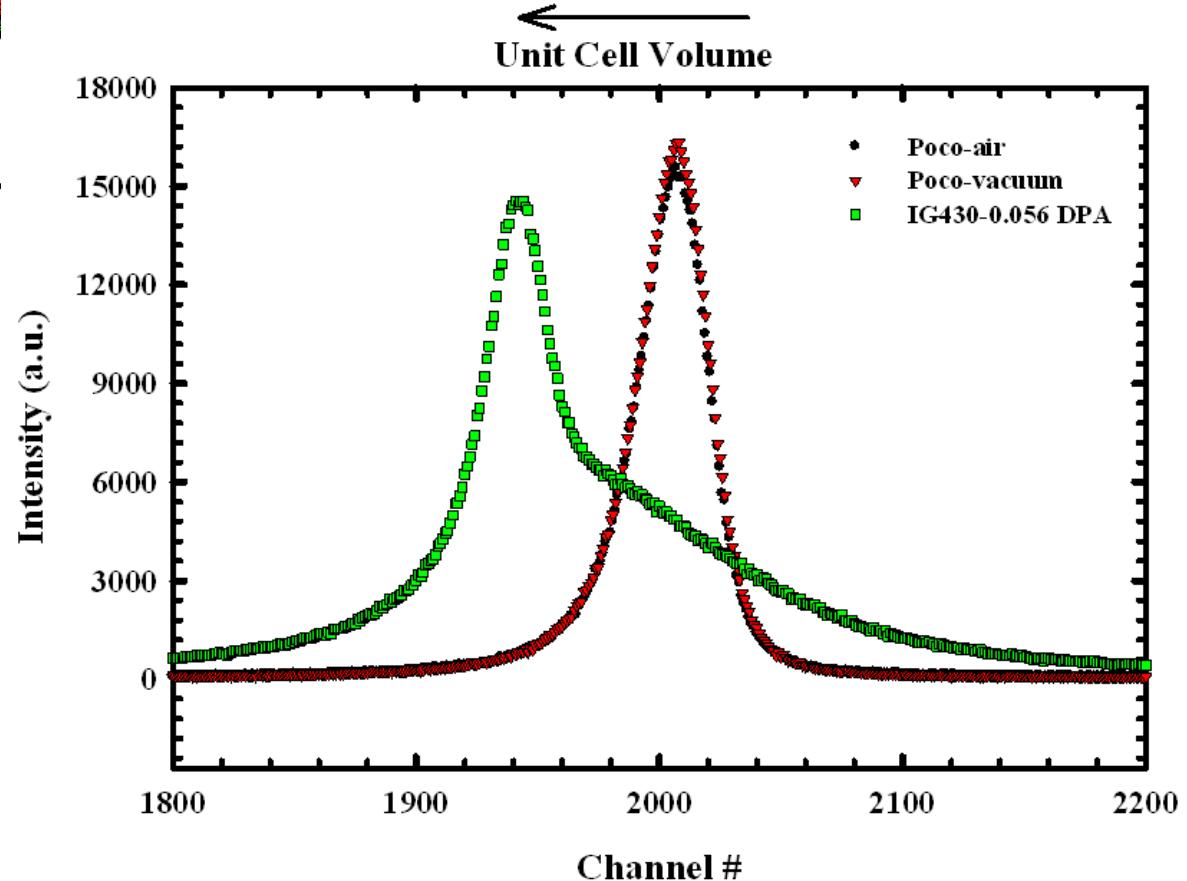
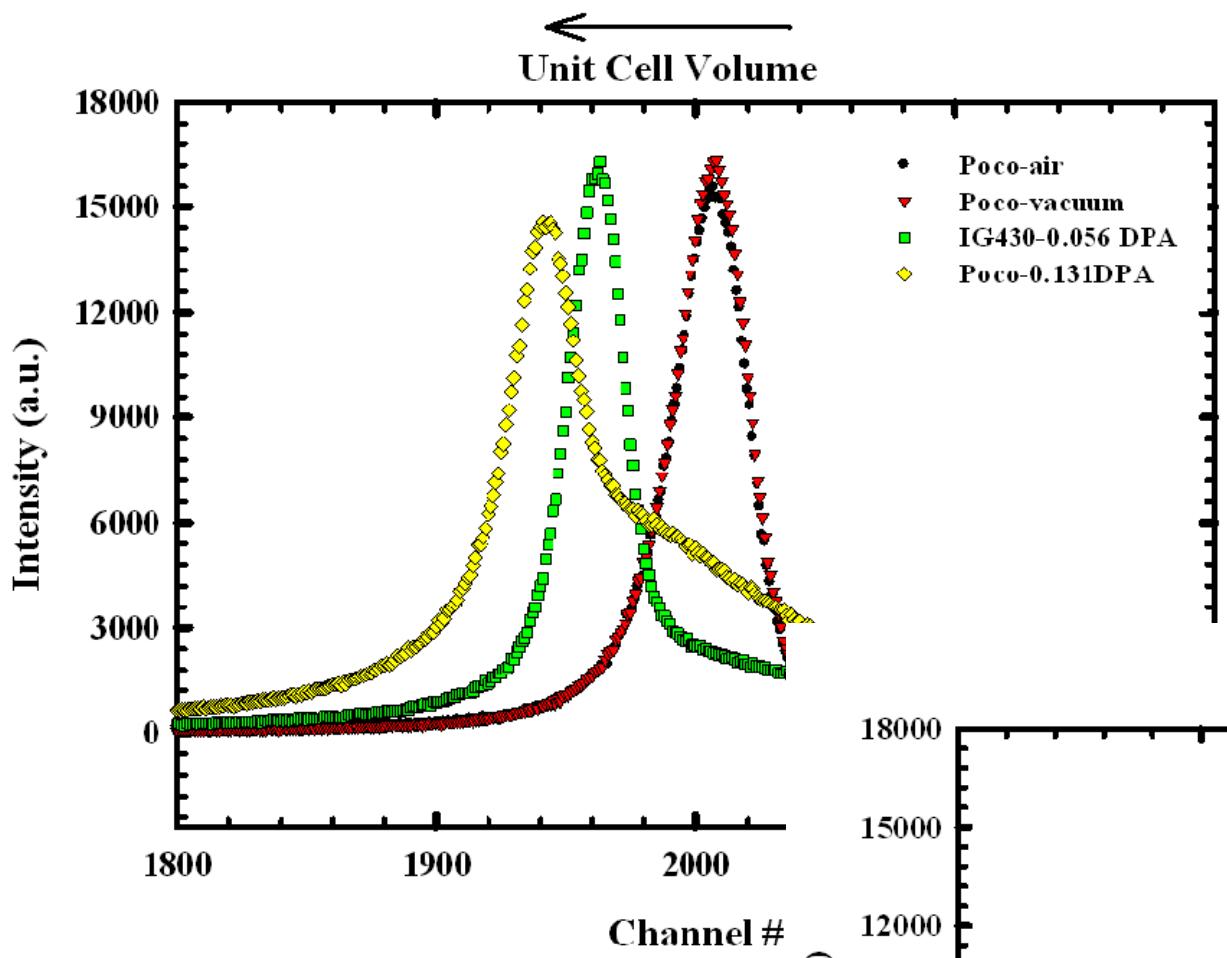


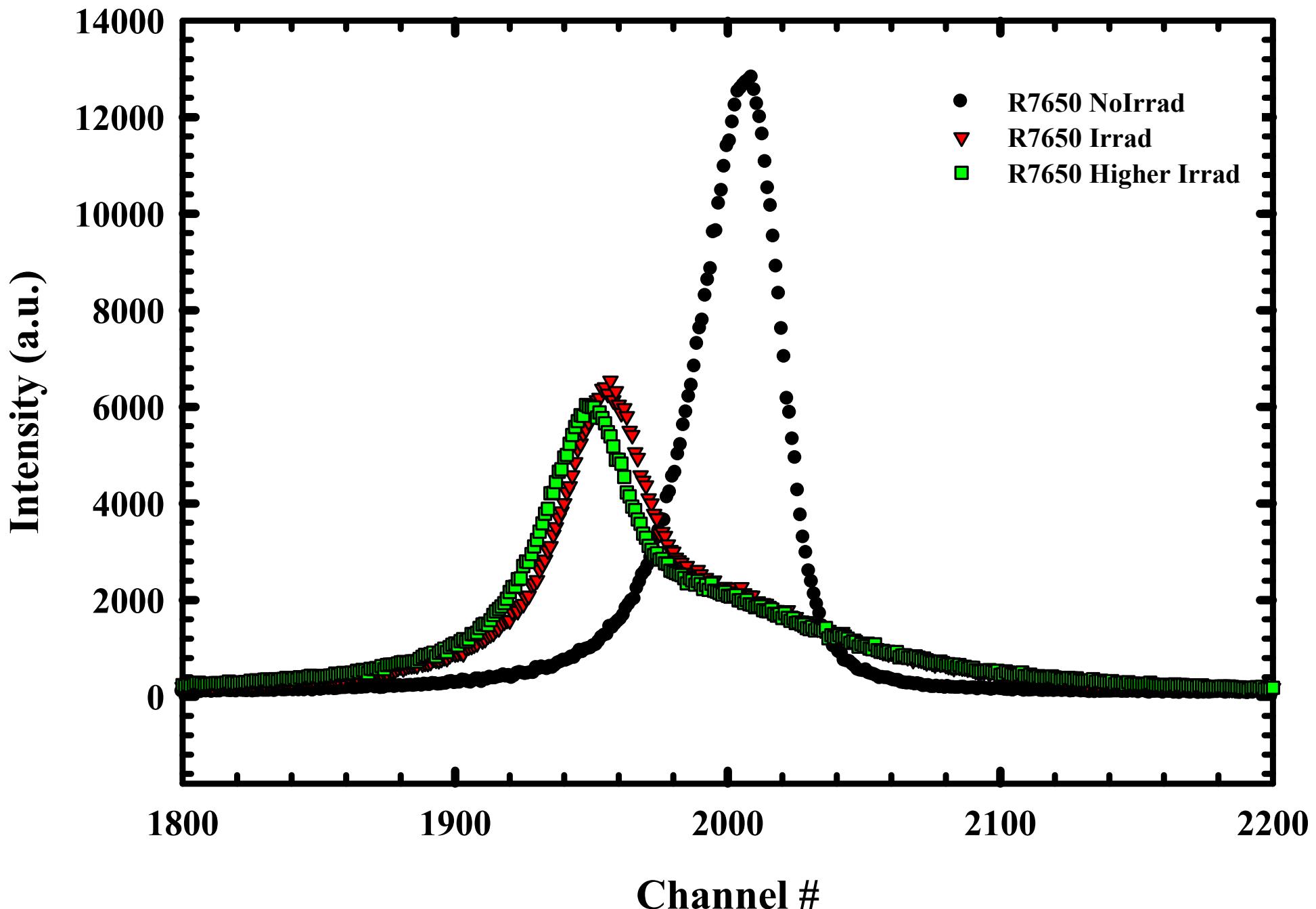
Goal is to correlate post-irradiation annealing observed macroscopically with shifts observed in XRD

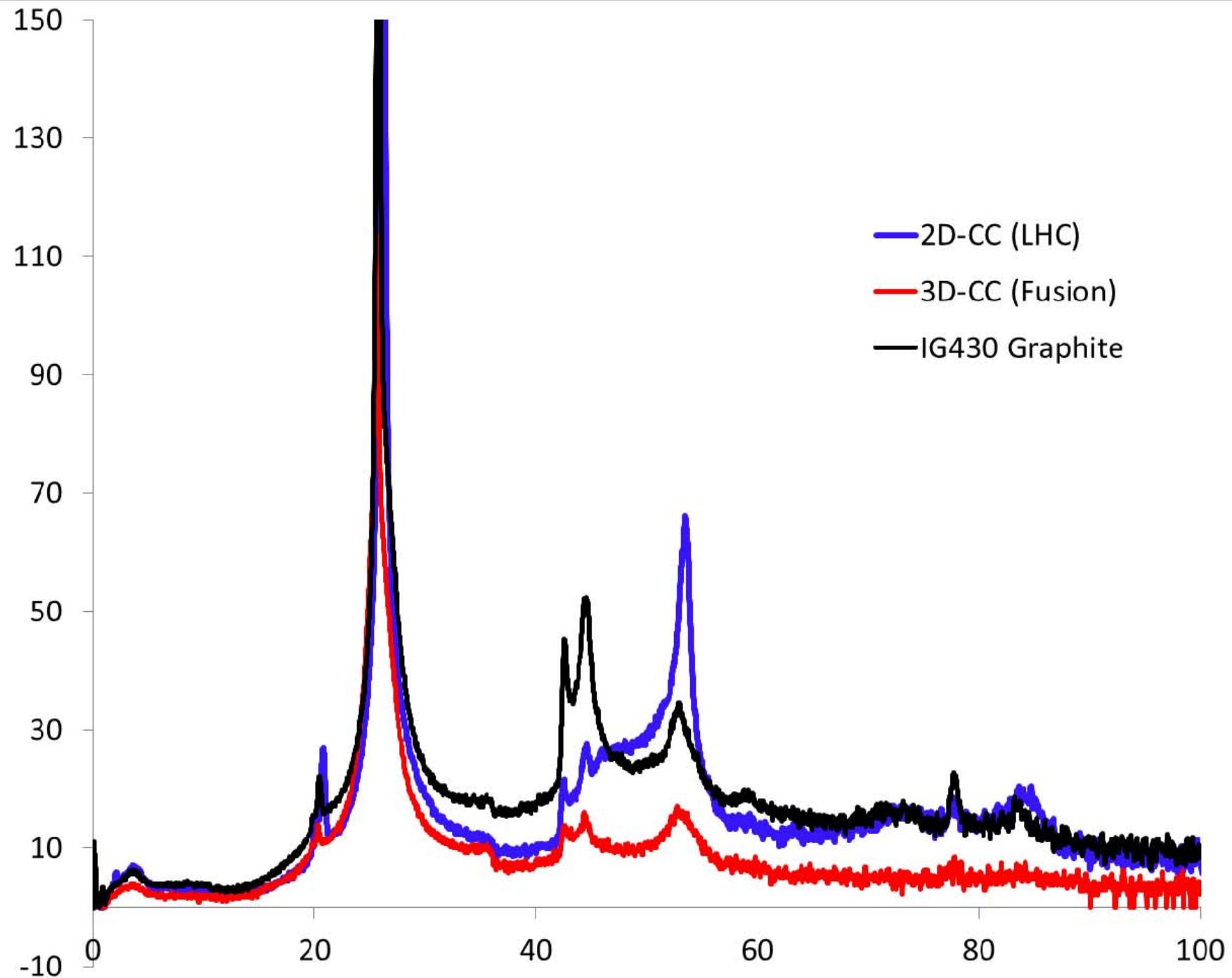
Global volumetric changes vs. crystal-level changes

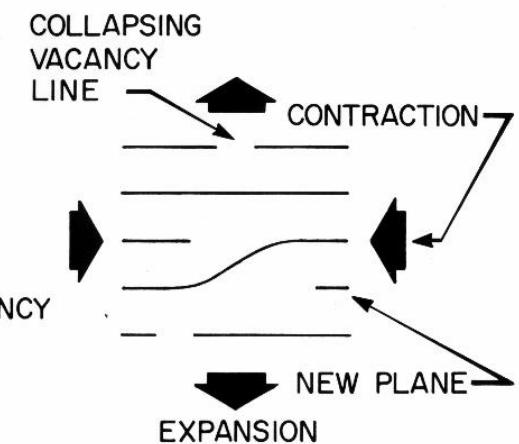
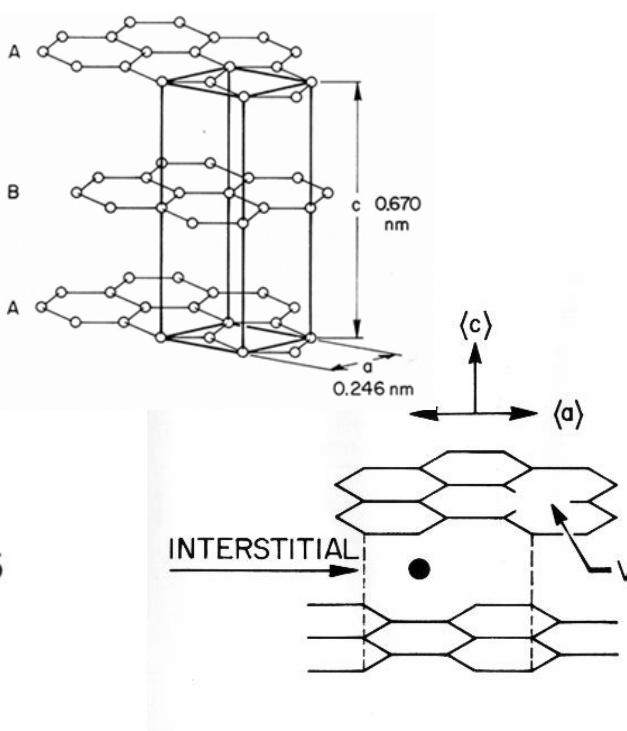
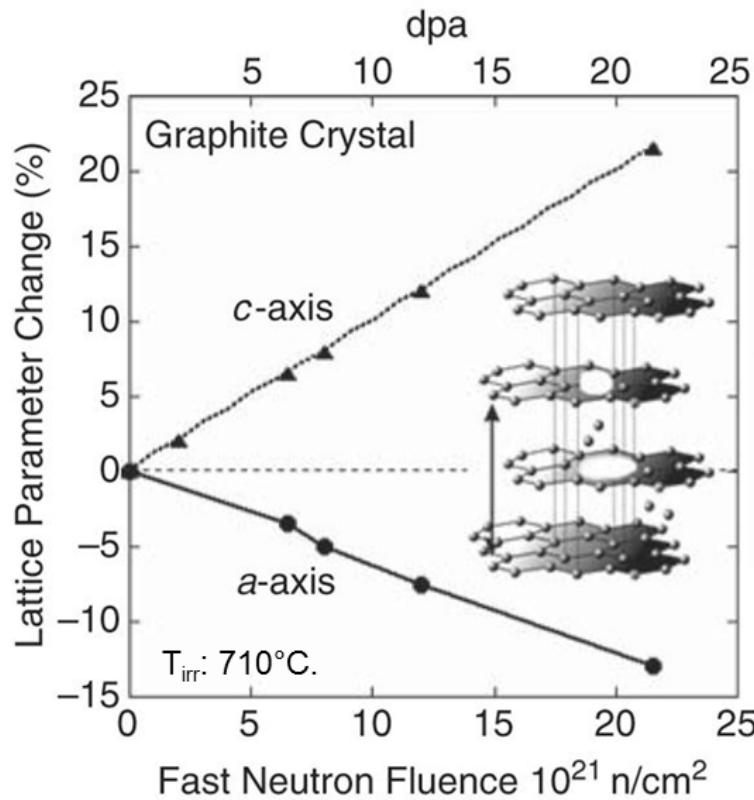
Activation Energy











Interstitial defects will cause crystallite **growth** perpendicular to the layer planes (c-axis direction)

Coalescence of vacancies will cause a **shrinkage** parallel to the layer planes (a-axis direction)